

METHOD FOR PRODUCING A REFRACTORY COMPOSITE MATERIAL

The invention relates to producing refractory composite materials, practically poreless, and can be used for production of composite articles with increased size stability, wear resistance, high specific physico-mechanical properties and hardness, production of wear-resistant inserts in components and materials for tribo-technical purposes as well.

A method for producing of refractory material is known, which consists of the following steps [1]: mixing of a pulverized refractory boride and/or a carbide with a carbon-containing substance; molding of article of predetermined shape from the mixture; heating of the prepared work-piece to extract carbon from the carbon-containing substance; introducing into the work-piece of a melted metal mixture comprising 75-99 %vol of at least one metal selected from the group containing Si, Cr, Fe, Ni, Ti and 1-25 %vol of a metal or mixture of metals from the group of Al, Cu, Fe and 0-24 %vol of the metal contained in the initial refractory material. The known method is quite complicated for realisation, as complicated equipment is necessary for precise dosing of molding, mixture components and melts used for infiltration of work-pieces and keeping of high temperatures. Because of significant shrinkage during sintering a closed porosity in the work-piece can be formed, thus resulting in deterioration of the subsequent metal infiltration into the porous work-piece.

Another method is also known for producing of refractory carbide-based composite articles of predetermined shapes [2]. According to the method a porous work-piece is molded with porosity 20-60 %vol from a pulverised carbide-forming metal, then it is heat-treated in a medium of gaseous hydrocarbon or a mixture of hydrocarbons at a temperature exceeding their decomposition temperature till an increase of the work-piece mass by at least 3 %, after which the prepared intermediate body is

infiltrated by a melt of metal from the following group of metals: Ag, Au, Cu, Ga, Ti, Ni, Fe, Co, or an alloy based on a metal from this group.

As carbide-forming metal at least one metal from groups IV, V or VI of the Periodic Table, for example Ti, Zr, Hf, V, Nb, Ta, Cr, Mo, W is used, while the heat-treatment is carried out in a medium of at least one hydrocarbon from the group including acetylene, methane, ethane, propane, pentane, hexane, benzene and their derivatives. As a hydrocarbon mixture for the heat treatment natural gas at temperature 750-950°C is used, for example.

Scope of properties of products prepared by the known method allows to apply them as refractory engineering materials, erosion-resistant electrodes for plasmotrons, high-current erosion-resistant electrical contacts, arc-extinguishing elements, high-temperature thermal accumulators, ablation heat-protective materials, damping refractory materials.

However, the known method is limited by a group of metals, which due to their physical properties can be introduced in composites, and in a series of cases it is necessary to prepare and use composite materials with metals in their composition, the properties of which make impossible to use the known method.

The object of the present invention is to develop a high-efficiency method providing an expansion of the range of metals possible to use as metallic phase of the composite; to prepare composite materials with high physico-mechanical characteristics, high hardness and wear resistance as well as workability at elevated temperatures, making it possible to change the above-mentioned properties by varying the ratio of components in composition both of the composite material and its metallic phase.

The essence of the invention is a method consisting of steps of infiltration of porous carbide work-piece with a metal, additional heat treatment in a melt of another metal at a temperature exceeding the melting point of the metal in intermediate body.

5 In particular, as the porous carbide work-piece it is possible to use a work-piece prepared by molding of carbide powders (for example  $\text{TiC}$ ,  $\text{B}_4\text{C}$ , etc.) and its subsequent sintering. Another variant is a porous work-piece formed of powders from carbide-forming elements or their mixtures with carbide powders with its subsequent treatment in a medium of hydrocarbons at a temperature exceeding their  
10 decomposition temperature followed by a treatment at a temperature of 1200-1800°C. The work-piece volume porosity can be either uniform, i.e. porosity in different parts of the work-piece is equal, or non-uniform, i.e. porosity in different parts of the work-piece is unequal, for example, if there is a porosity gradient in one or several directions.

15 As a precursor in the method not only the stated types of porous carbide work-pieces can be used but also work-pieces prepared by other methods. The condition for a porous carbide work-piece is its porosity to be in the range of 30-60 % vol. The major part of pores should be open in this case. Using of work-pieces of porosity  
20 beyond this range is not expedient as it results in deterioration of the properties of composite materials (at high porosity) or complicates the production process of the intermediate body (at low porosity).

The porous work-piece is infiltrated by a melt of metal or its alloy resulting in  
25 preparation of an intermediate body. The infiltration is carried out in inert medium by dipping of the work-piece in a melt or by melting of a weighed sample of metal or alloy on surface of the work-piece.

Further, the prepared intermediate body is additionally treated in a melt of another  
30 metal, the temperature of which is higher than the melting point of the metal phase

of the intermediate body. It is more preferable to use such metals, which do not strongly (chemically) interact with the carbide phase of the material (thus, the carbide skeleton is kept during the production process, and, hence the strength of the prepared material) and also able to form a common liquid phase with the metal of the intermediate body in melts.

For acceleration of the process in a series of cases before the treatment in a melt the intermediate body is heated up to a temperature exceeding the melting point of the metal phase of the intermediate body.

The composite material prepared by the claimed method represents a two-phase system formed by a continuous three-dimensional skeleton of refractory carbide phase with pores filled by metal phase. In this case metal phase is a mixture of metals, namely initial one and a metal from the melt. Depending on the treatment conditions in a melt (temperature of melt, treatment time, fluidity of melt, etc) the metal from the melt can be distributed either uniformly or non-uniformly throughout the volume of the composite material.

The essence of the present invention is in that the intermediate body comprising a carbide skeleton and a metal is treated in a melt of another metal at a temperature exceeding the melting point of the metal in the intermediate body. In this case, the metal in pores of the carbide skeleton is in a liquid state. During the interaction with the melt of another metal a process of mutual diffusion of the metal from the intermediate body into the melt takes place and in a reverse direction – from the melt into the intermediate body. This results in replacing of the metal in composition of the intermediate body by the metal from the melt and the degree of replacement by the new metal may reach 80 % by volume of the metallic phase of the composite material and greater. It should be noted that it is not a decisive importance in this invention whether the metal from the melt infiltrates the carbide skeleton or not at the temperature of treatment in the melt. The metal, by which the intermediate body

was infiltrated, provides a reliable adhesion bonding in the interface of carbide and metal phases in a ready-made material.

In case of a non-uniform distribution of metal from the melt in the composite material volume, for example, if there is a replacing metal on surface and is absent inside the composite material, this offers possibilities of production of articles with a gradient by composition. The latter provides the articles with a complex of unique properties, for example, high strength and rigidity of the composite as a whole and anti-friction properties due to the near-surface zone enriched by another metal.

The conditions for producing of a refractory composite material are different in each case and depend on the choice of:

- initial porous work-piece,
- metal or alloy for infiltration of porous work-piece and producing of intermediate
- body,
- metal of melt for treatment of intermediate body,
- application of article.

Essence of the invention is disclosed in the following examples.

*Example 1.* The initial porous work-pieces are shaped as discs of sizes Ø20x5 mm. The preparation process of the work-pieces is as follows. A mixture of powders of amorphous boron (69%wt) and boron carbide (29 %wt) with addition of temporary bond (phenol-formaldehyde resin SF 10-A – 2% wt) are molded to form discs of sizes Ø20x5 mm. Further they are placed in an isothermal reactor for pyrocarbon synthesis and are heat-treated at temperature of 870°C in a medium of natural gas until the mass is increased by 19 %. The treated discs are placed in a vacuum furnace and are heated up to temperature 1620°C and are kept there at this temperature during 20 min. The prepared work-pieces have porosity of 50 % vol, which is equally distributed throughout the volume. The work-pieces are infiltrated at temperature 1200-1250°C in the vacuum furnace by melting of weighed samples

of aluminium of technical purity mark A7 ( $Al \geq 99,7\%$ ) on surface of the work-pieces. Further the intermediate bodies  $B_4C/Al$  are heated up to  $700^\circ C$  and are immersed in Zn melt heated up to  $750^\circ C$  and are kept there for 2 hours. As a result the samples containing 50 % vol of boron carbide and 50 % vol of the alloy (Al-Zn) are produced. The metallic phase contains 54 % vol of aluminium and 46 % vol of zinc. The material has the following properties: density –  $3.57\text{ g/cm}^3$ ; hardness – HRA 57.

*Example 2.* The initial porous work-pieces are shaped as discs of sizes  $\varnothing 20 \times 5\text{ mm}$ . The preparation process of the work-pieces is as follows. A mixture of powders of titanium (49 % wt) and titanium carbide (49 % wt) with addition of temporary bond (phenol-formaldehyde resin SF 10-A – 2 % wt) are molded to form discs of sizes  $\varnothing 20 \times 5\text{ mm}$ . Further they are placed in an isothermal reactor for pyrocarbon synthesis and are heat-treated at temperature of  $870^\circ C$  in a medium of natural gas until the mass is increased by 12.5 %. The treated discs are placed in a vacuum furnace and are heated up to temperature  $1700^\circ C$  and are kept there at this temperature during 20 min. The prepared work-pieces have porosity of 40 % vol, which is equally distributed through the volume. The work-pieces are infiltrated at temperature  $1150\text{--}1200^\circ C$  in the vacuum furnace by melting of weighed samples of aluminium of technical purity mark A7 ( $Al \geq 99,7\%$ ) on surface of the work-pieces. Further the intermediate bodies  $TiC/A7$  are heated up to  $700^\circ C$  and are immersed in Mg melt heated up to  $750^\circ C$  and are kept there for 4 hours. As a result samples containing 60 % vol of titanium carbide and 40 % vol of the alloy (Mg-Al) are produced. The metallic phase contains 78 % vol of magnesium and 22 % vol of aluminium. The material has the following properties: density –  $3.63\text{ g/cm}^3$ ; bending strength – 425 Mpa.

Properties of the material were determined using the methods listed below:

1. Density was determined by hydrostatic method
2. Hardness by Rockwell method
3. Bending strength – by three-point bending method.